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Tunable mesoporous $g - C_3N_4$ nanosheets as a metal-free catalyst for enhanced visible-light-driven photocatalytic reduction of U(VI)

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HIGHLIGHTS

- Tunable mesoporous in mpg-C₃N₄ improved the photocatalytic reduction for U(VI).
- \bullet mpg-C₃N₄ exhibited a high photocatalytic reduction efficiency to U(VI).
- The photocatalytic reduction of U(VI) was highly selective to coexisting ions.
- The photoconversion process of U(VI) was addressed using XPS, XRD and EXAFS.

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GRAPHICAL ABSTRACT

ABSTRACT

Although photocatalytic reduction has been proven to be a green, efficient, and economical strategy for uranium (VI) extraction from water, it faces several challenges, such as low quantum conversion efficiency and utilization of sunlight. In this work, the photocatalytic efficiency of $g-C_3N_4$ was tuned by introducing proportionally-adjustable mesoporous. The prepared mesoporous g-C3N4 samples (MCN*r*, *r* represents the initial silica/cyanamide mass ratios during synthesis) exhibited higher surface area and larger pore volume. Compared with pristine g- C_3N_4 (BCN), both light utilization and photo-generated carrier separation efficiency were significantly improved for MCN*r*. MCN*^r* showed an enhanced photocatalytic performance for the reduction of U(VI) under visible light. The optimum photocatalytic performance was achieved for $MCN_{1.0}$, which was 6.75 times higher than that of BCN. Both transmission electron microscopy (TEM) and X-ray diffraction (XRD) have confirmed the formation of $UO₂$ on MCN_{1.0} surface after light irradiation. X-ray photoelectron spectroscopy (XPS) and extended X-ray absorption fine structure (EXAFS) further revealed the re-oxidation of UO₂ during the photocatalytic process. A high uranium extraction capacity (\sim 2990 mg/g) could be achieved by MCN_{1.0}, and the deposited UO₂ could be easily eluted by 0.01 mol/L Na₂CO₃ solution after exposure to air, showing high reuse performance. In the presence of co-existing ions, the photocatalytic reduction of U(VI) remained a high selectivity.

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1. Introduction

As the fast development of nuclear energy, the uranium resource has become increasingly crucial to the nuclear industry [\[1\].](#page-9-0) However, the terrestrial supply of uranium is limited, which was reported to be depleted in one century $[1,2]$. As a result, seawater uranium extraction has attracted extensive attention for a long time, owing to the large content of uranium in seawater (~4.5 billion tons), which is approximately a thousand times greater than terrestrial uranium resources [\[3\]](#page-9-1). In the past 50 years, adsorption strategy has been widely used for uranium extraction from seawater, mostly using functionalized polymer fibers as sorbents [\[4–11\]](#page-9-2). However, the extraction of uranium from seawater via physicochemical adsorption is significantly challenging, mainly owing to the extremely low U(VI) concentrations $(\sim 3.0 \text{ pb})$ and the abundant coexisting metal ions (e.g., Na⁺, K⁺, Mg²⁺, Ca²⁺, Cu^{2+} , Fe³⁺, VO₂⁺) [\[1,11\].](#page-9-0) Moreover, the elution conditions for captured U(VI) and V(V) are generally harsh enough to destroy the sorbents, making the sorbents hard to be reused [\[12\].](#page-9-3) Thus, there is an urgent need to develop new strategies for seawater uranium extraction.

Our previous work showed that efficient seawater uranium extraction could be achieved by applying a photocatalysis method [\[13,14\]](#page-9-4), where U(VI) was reduced to U(IV), existing in the form of $UO₂$ nanoparticles on TiO₂ surface. Compared with physicochemical adsorption, the photocatalysis method has higher efficiency and selectivity [\[15–19\]](#page-9-5), and has been proven to be a non-polluting, economical, and facile method for U(VI) extraction [\[13,20–25\]](#page-9-4). Despite the remarkable advantages, proper photocatalysts are required for this technique. To date, TiO₂ has been mostly applied in the photocatalytic U(VI) reduction, where UV irradiation was necessary due to the wide bandgap of TiO2 (3.2 eV). However, UV light only occupies about 4% of solar energy, which seriously inhibits the photocatalytic efficiency and practical applications [\[26\].](#page-9-6)

In the past decade, metal-free graphitic carbon nitride $(g-C_3N_4)$ has attracted extensive attentions owing to its low cost, high stability, relative facile synthesis, and moderate energy gap (-2.7 eV) [\[27\].](#page-9-7) These unique properties enable $g-C_3N_4$ to be widely used for visible light photocatalytic applications such as water splitting, $CO₂$ reduction, organics photodegradation, and fuel cells [\[28–35\]](#page-9-8). The conduction band (CB) position of g-C₃N₄ is -1.23 V (vs. SHE at pH 7.0), which is more negative than the reduction potential of $UO_2^2^+ / U^{4+}$ (0.267 V) and UO_2^2 ⁺/UO₂ (0.411 V) [\[24,25\]](#page-9-9). Accordingly, the photocatalytic U(VI) reduction is thermodynamically feasible. However, bulk $g - C_3N_4$ (BCN) obtained through the direct thermal polymerization process generally yields relatively low specific areas and less active sites, which leads to poor performance in catalytic processes [\[33\].](#page-9-10) In addition, the quantum efficiency of BCN in the visible light region needs further improvement. It has been shown that the electronic band structures of $g - C_3N_4$ could be tuned by the modification of morphology [\[36\].](#page-9-11) The fabrication of mesoporous-structured $g-C_3N_4$ is an effective method to promote the photocatalytic performance of g-C3N4 without introducing any impurity [\[37,38\].](#page-10-0) Compared with bulk samples, porous $g-C_3N_4$ has a higher specific surface area and accessible porosity, which can provide more active chemical sites for photocatalytic reactions as well as higher sizeor shape-selectivity [\[39\].](#page-10-1) This can be attributed to the fact that mesoporous structures can enhance light utilization through increasing light scattering and refraction processes in the pore channels [\[38,40,41\].](#page-10-2)

Considering the excellent physicochemical and optical properties of mesoporous $g - C_3N_4$, it is expected to be a good candidate to efficiently extract U(VI) from water. Therefore, mesoporous g-C₃N₄ (MCN_r) samples with different amounts of pores were fabricated and characterized in this study. The photocatalytic performance and mechanism of U(VI) reduction over the synthesized samples were evaluated and addressed in detail combining spectroscopic approaches. The results will serve to enhance understanding of the photocatalytic reduction of U(VI), and are promising to promote the potential application of the photocatalytic method under visible light.

2. Experimental section

2.1. Photocatalysts preparation

[Scheme 1](#page-1-0) illustrates the synthesization of mpg- C_3N_4 with the hard template method [\[41\].](#page-10-3) Briefly, 1.0 g of cyanamide (99%) was stirred at 70 °C, and different amounts (0.625, 1.25, 2.5, 5.0, and 10 g, respectively) of 40% dispersion of $SiO₂$ particles with a size of 12 nm (Ludox HS40, Aldrich) were added dropwise. The mixtures were heated overnight at 70 °C with stirring to evaporate water. Afterwards, the dried sample was thermally treated at 550 °C for 4 h in a tube furnace under N_2 atmosphere, where the heating rate was $2 °C/min$. Subsequently, the reactor was spontaneously cooled to 25 °C. The resulting yellow powders were treated with 4.0 mol/L NH₄HF₂ for 24 h to remove the $SiO₂$ template, and then was washed with distilled water and ethanol for several times. Finally, the powders were dried with a vacuum freeze dryer. The as-prepared samples were labeled as MCN*^r* (*r* refers to the initial silica/cyanamide mass ratios of 0.25, 0.5, 1.0, 2.0 and 4.0, respectively). For comparison, a bulk $g-C_3N_4$ sample (labeled as BCN) was synthesized by directly heating 1.0 g of cyanamide in the tube furnace at 550 °C.

2.2. Characterization

Morphology and structures were characterized with transmission electron microscopy (TEM, Fei Tecnai G2 F30) and X-ray diffraction (XRD) (D/Max-2400, Rigaku, Japan) instrument equipped with Cu Kα, respectively. Fourier-transform infrared (FT-IR) spectra were obtained on a Bruker Alpha instrument, and the spectra were average over 120 scans with a spectral resolution of 4 cm^{-1} . The Brunauer-Emmett-Teller (BET) area was measured by Tristar II 3020, and degassing was performed at 100 °C before detection. Pore-size distributions were calculated with the Barrett–Joyner–Halenda (BJH) method. The surface chemical component and oxidation states were characterized using Xray photoelectron spectroscopy (XPS) (Kratos AXIS Ultra^{DLD}) with Al-Kα radiation at 10 kV and 5 mA. The binding energies of XPS spectra were referenced by C 1 s at 284.6 eV. UV–vis diffuse reflection spectrum (DRS) was gained using Kubelka-Munk function $(F(R_\infty) = (1-R_\infty)^2$ / (2*R*∞)) by UV–vis spectrophotometer (UV-2550, Shimadzu, Japan),

Scheme 1. Schematic diagram for the preparation of MCN*^r* samples.

Fig. 1. XRD patterns (A); FT-IR spectra (B); TEM images for BCN (C), $MCN_{0.25}$ (D), $MCN_{0.5}$ (E), $MCN_{1.0}$ (F), $MCN_{2.0}$ (G), and $MCN_{4.0}$ (H).

where R_{∞} was the reflectance of samples. According to the equation: $(\alpha h v)^{1/2} \propto hv - Eg$, the bandgap energy (Eg) was obtained by extrapolating the linear portion of the plots between (α*hν*) 1/2 against *h*ν to Yaxis = 0, where α was the absorption coefficient, ν is the frequency, and *h* is Planck's constant [\[42\]](#page-10-4). Photoluminescence (PL) spectra and the photoluminescence emission lifetimes were obtained through a fluorescence spectrophotometer (FLS920, Edinburgh Instrument, UK) at room temperature. The electron spin resonance (ESR) measurement was carried out on Bruker A300 spectrometer under visible light irradiation $(\lambda > 420 \text{ nm})$, where 5,5-dimethyl-l-pyrroline N-oxide (DMPO) was used as a probe.

Uranium L_{III} -edge X-ray absorption near edge structure (XANES) and extended X-ray absorption fine structure (EXAFS) were collected at the beamline BL14W1 station of Shanghai Synchrotron Radiation Facility (SSRF, China). The electron beam energy and average stored current were 3.5 GeV and 220 mA, respectively. During the measurements, the double-crystal Si (111) monochromator was slightly detuned to reduce the high-order harmonics and to control the total fluorescence counts. XANES and EXAFS were collected under fluorescence mode with a Lytle detector [\[43\].](#page-10-5) The EXAFS data were analyzed using Athena and Artemis interfaces compacted in IFFEFIT [\[44\]](#page-10-6) and FEFF 7.0 [\[45\]](#page-10-7).

2.3. Photoelectric chemistry experiments

The photocurrent was measured on a CHI 660D electrochemical workstation (Shanghai Chenhua, China) in a typical three-electrode potentiostat system. Briefly, 0.7 mg photocatalyst and 0.3 mg graphitized carbon black were added into a 0.95 mL ethyl alcohol and 50 μL Nafion solution. The formed slurry was then dip-coated onto the L-type glassy carbon electrode, which was used as the working electrode after drying at 40 °C for 30 min. A Pt sheet and standard Ag/AgCl (in saturated KCl) were used as the counter electrode and reference electrode, respectively. For the electrolyte, 0.1 mol/L of Na₂SO₄ aqueous solution was used. The variations of photoinduced current density with time (*i*–*t* curve) were measured using a Xe lamp (300 W) with a UV-cutoff filter as the light resource ($\lambda > 420$ nm).

2.4. Photocatalysis tests

In the catalytic process, 75 mg MCN*^r* was added into 15 mL 1.0×10^{-4} mol/L UO₂²⁺ solution (containing 2.5 mL methanol as the electron sacrifice). The pH value was adjusted to 6.0 with a negligible volume of HCl and/or NaOH solution. The visible light irradiation was obtained from a 300 W Xe lamp equipped with a 420 nm cutoff filter. Before irradiation, the reaction system was bubbled with argon in dark for 2 h to maintain anaerobic conditions and reach the adsorption–desorption equilibrium. After irradiation, 1 mL of suspension was pipetted out at a certain time and rapidly filtered using a $0.22 \,\mu m$ membrane. U(VI) concentration was measured by UV–vis spectrophotometry at 652 nm using Arsenazo III (The detection limit for uranium is lower than 1.0×10^{-7} mol/L) and inductively coupled plasma optical emission spectrometry (ICP-OES, the detection limit for uranium was \sim 100 ppb). UO₂²⁺ solution without photocatalysts was also irradiated under the above conditions for comparison. The concentrations of metal ions (Cu, Fe, V, Mg, Ca, Na, K, and Zn) were detected by ICP-OES. In the quenching experiments, tertiary butanol (TBA), p-benzoquinone (P-BQ) and methanol were added as scavengers for 'OH radicals, O_2 ⁻ radicals, and holes, respectively [\[22\]](#page-9-12).

3. Results and discussion

3.1. Characterization

The structures of the as-prepared MCN_r ($MCN_{0.25}$, $MCN_{0.5}$, $MCN_{1.0}$, $MCN_{2.0}$, and $MCN_{4.0}$) and BCN samples were studied by XRD as shown

Fig. 2. N₂ adsorption-desorption isotherms at 100 °C (A); The corresponding PSD curves (B); Photograph of 200 mg powder and dispersion in water (C); XPS spectra of BCN and $MCN_{1.0}$ (D); The corresponding high-resolution C1s spectra (E) and N1s spectra (F); UV–vis DRS of BCN and MCN*^r* photocatalysts (G); Plot of $(Ah\nu)^{1/2}$ versus photon energy (eV) (H); *I-t* spectra of as-synthesized BCN and MCN*^r* samples (1: $MCN_{1.0}$; 2: $MCN_{2.0}$; 3: $MCN_{4.0}$; 4: $MCN_{0.5}$; 5: $MCN_{0.25}$; and 6: BCN) (I).

in [Fig. 1](#page-2-0)A. For both MCN*^r* and BCN, the peaks at 13.1° (assigned to $(1 0 0)$ faces), 17.8° (assigned to $(6 0 0)$ faces), and 27.4° (assigned to (0 0 2) faces) were observed in the XRD pattern, which can be ascribed to the in-plane structural packing of tri-s-triazine motifs, the denser packing or a distortion of the melon structure, as well as the inter-layer stacking of conjugated aromatic systems, respectively [\[24,46\].](#page-9-9) Nevertheless, peak broadening related to the (0 0 2) faces and a decrease in the overall peak intensity were observed for $MCN_{2,0}$ and $MCN_{4,0}$, implying lower crystallinity. Such phenomena are possibly due to [\(1\)](#page-4-0) incomplete polymerization and structural defects, reflecting the effect of geometric confinement in the nanosized pore channel [\[41\];](#page-10-3) and [\(2\)](#page-8-0) the formation of small $g-C_3N_4$ pieces when excess templates was used. The functional groups on MCN*^r* were characterized by FT-IR, as shown in [Fig. 1](#page-2-0)B. The broad absorption band at ~2900–3600 cm−1 could be ascribed to the stretching vibration modes for the N–H and hydroxyl groups of adsorbed H_2O . The peaks in the range from 1200 to 1700 cm^{-1} represent the stretching modes of aromatic C₃N₄ heterocycles, while the peak at ~809 cm^{$^{-1}$} could be attributed to the typical breathing mode of the triazine units [\[47\].](#page-10-8) All the MCN*^r* samples (MCN_{0.25}, MCN_{0.5}, MCN_{1.0}, MCN_{2.0}, and MCN_{4.0}) exhibited characteristics similar to BCN, which suggests a similar chemical structure. It is noteworthy that the intensity of the band at 2900–3600 cm−1 of the MCN_r samples is quite strong in contrast to BCN, especially in $MCN_{1.0}$. The result implies the existence of the porous, which could adsorb a larger amount of $H₂O$ than the bulk one [\[37\].](#page-10-0)

[Fig. 1](#page-2-0)C–H and Fig. S1 (in Supplementary Information, SI) show the morphology and microstructure of BCN and MCN*r*. It is clear that all samples were laminar. Compared with BCN, wormhole-like mesostructures without long-range order could be observed on MCN*r*. For $MCN_{0.25}$, few wormhole structures were disorderly distributed on the bulk surface. As the template amount increased, more wormhole-like mesostructures appeared and were uniformly distributed. However, the

mesoporous structure were almost destroyed when the mass ratio of templates to cyanamide increased to 4.0 ($MCN_{4.0}$), and an amount of small-sized plates were observed to stack together. This was considered that, with the large amount of templates, the in-plane structural packing of tri-s-triazine motifs was interrupted to some extent. It was also supported by the XRD patterns ([Fig. 1A](#page-2-0)), where the intensity of $(1 0 0)$ faces significantly decreased. In the case of MCN_{4.0}, the BCN layers were completely "cut" into small and irregular pieces by templates (20–30 nm, Fig. S2). These pieces clustered together, followed by the increase to some extent in the thickness [\(Fig. 1H](#page-2-0)) [\[37,41\]](#page-10-0). The formation of small pieces of $MCN_{4.0}$ further supported the observation of the broaden (0 0 2) peaks in the XRD patterns.

[Fig. 2A](#page-3-0) displays the nitrogen adsorption-desorption isotherms of the MCN_r and BCN. In the case of pristine BCN, a weak N_2 adsorption possibly suggested the nonporous structure. While all MCN*^r* samples demonstrated the type-IV isotherms bearing H1-type hysteresis loops, indicating mesoporous characteristics of the prepared photocatalysts [\[47\]](#page-10-8). To further analyze the pore structures of MCN*^r* samples, the pore size distributions (PSD) were calculated using the BJH method from the desorption branch of the nitrogen isotherms [\(Fig. 2B](#page-3-0)). The pristine g- C_3N_4 showed negligible pores, and the average pore sizes were uniformly near 12 nm for the $MCN_{0.25}$, $MCN_{0.5}$, and $MCN_{1.0}$ samples, which could be regarded as a replica of the structure of the silica spheres (\sim 12 nm, see Fig. S1A). However, for MCN_{2.0} and MCN_{4.0}, a significant decrease in the number of pores was demonstrated. The small pore size further confirmed the speculation that stackable small C_3N_4 pieces could be formed when the content of templates was high.

BET specific surface area (SSA), pore size, and average pore volume are listed in [Table 1.](#page-4-1) Compared with BCN $(9 \text{ m}^2/\text{g})$, the SSA of MCN_{*r*} were significantly improved (23–191 m²/g), depending on the content of template. $MCN_{1,0}$ had the largest surface area among all the asprepared samples. $MCN_{0.25}$ and $MCN_{0.5}$ possessed fewer holes

Table 1 The structure parameters of BCN and MCN*^r* samples.

sample	BET surface area (m^2/g)	Total pore volume $\text{cm}^3\text{/g}$)	Average pore diameter (nm)	Band-gap (eV)
BCN	9.1 ± 0.8	0.04	17.6	2.73
$MCN_{0.25}$	34.1 ± 1.1	0.10	11.9	2.81
MCN _{0.5}	$111.0 + 7.1$	0.31	11.1	2.79
MCN ₁₀	$190.7 + 10.0$	0.52	10.9	2.92
MCN ₂₀	$124.5 + 6.2$	0.32	10.1	2.94
MCN_{40}	23.0 ± 2.5	0.05	7.5	2.84

compared with $MCN_{1.0}$, owing to the lower content of template. In the cases of $MCN_{2.0}$ and $MCN_{4.0}$, the limited pores resulted in smaller SSA. The SSA and total pore volume of $MCN_{1.0}$ were \sim 191 m²/g and \sim 0.52 cm³/g, respectively, which were 21.8 and 13.0 fold higher than that of BCN, respectively [\(Table 1\)](#page-4-1). Compared with BCN, the high SSA and large pore volume for $MCN_{1.0}$ resulted in a much larger volume and better water dispersibility [\(Fig. 2](#page-3-0)C) [\[47\].](#page-10-8)

To further explore the surface species and elemental valence states in MCN*^r* samples, the XPS measurements were conducted. As shown in [Fig. 2D](#page-3-0), XPS spectrum of $MCN_{1.0}$ mainly consisted of three sharp peaks at 287.3, 398.3, and 531.3 eV, which were ascribed to C 1 s, N 1 s, and O 1 s signals, respectively. The O 1 s signal may be due to the surface absorbed CO_2 and H₂O [\[47\].](#page-10-8) To gain insight into the chemical bonding between C and N atoms, high-resolution spectra of C 1 s and N 1 s were analyzed, respectively ([Fig. 2E](#page-3-0) and F). For $MCN_{1.0}$, the C 1 s spectrum was composed of sp² C—C, C=N/C≡N (from the defect-containing sp²hybridized carbon atoms), and N=C-N bonds at 284.6 eV , 286.0 eV , and 288.0 eV, respectively [\(Fig. 2E](#page-3-0)). The N 1 s spectrum ([Fig. 2](#page-3-0)F) was deconvoluted into N=C-N (~398.5 eV), N-(C)₃ (~399.8 eV), N-H (-400.8 eV) , and C-N-H (~404.3 eV, from the positive charge localization in heptazine rings due to incomplete condensation), respectively [\[48\].](#page-10-9) In addition, it was noted that the surface C/N atomic ratio of $MCN_{1.0}$ (0.57) slightly decreased relative to BCN (0.53), which may be attributed to uncondensed amino functions [\[41\]](#page-10-3). Therefore, it was speculated that the structural defects can selectively create more carbon vacancies (CVs), which are also expected to improve the photocatalytic activity of g- C_3N_4 [\[49\].](#page-10-10)

The optical absorption properties of BCN and MCN*^r* were also measured by UV/vis DRS. As shown in [Fig. 2G](#page-3-0), all samples demonstrated a typical semiconductor absorption pattern in the visible range. Compared to BCN, an upshift in light absorbance in the region of 400–600 nm was observed for all MCN*^r* samples. The enhanced lighttrapping effect was attributed to multiple reflection or transmission of light scattered by the pore walls in the MCN*^r* body [\[38\].](#page-10-2) The high visible light absorbance indicates that the obtained MCN*^r* samples have higher visible light utilization efficiency, which is conducive to enhance the photoactivity. In addition, an increase in the band gap energy was observed for MCN*^r* ([Fig. 2H](#page-3-0) and [Table 1](#page-4-1)), which was caused by the quantum confinement effect with the CB and VB shifting in opposite directions [\[38,47,50\]](#page-10-2).

The photocurrent-time measurement in on-off cycles was employed to evaluate the photo-carriers separation dynamics of BCN and MCN*^r* electrodes under visible light irradiation. Compared with BCN, MCN*^r* photocatalysts displayed much higher photocurrent intensity following in the sequence of $MCN_{1.0} > MCN_{2.0} > MCN_{4.0} > MCN_{0.5}$ $>$ MCN_{0.25} $>$ BCN [\(Fig. 2](#page-3-0)I). Obviously, both the largely increased specific surface area and abundant pore walls lead to the enhancement of visible light absorption, although the bandgap was increased for MCN_r. The results further indicated that the introduction of mesoporous played a critical role in charge separation and the increased visible light absorption. The photocurrent intensity generated by the $MCN_{1.0}$ electrode was the highest (4.11 μA cm⁻²), which was 4.0 times of that

induced by BCN (1.03 μA cm⁻²). As a result, MCN_{1.0} exhibited the highest efficiency in separating photo-generated charge carriers [\[51\].](#page-10-11)

3.2. Photocatalytic activity under visible light

The photocatalytic activities of BCN and MCN*^r* for U(VI) reduction are shown in [Fig. 3A](#page-5-0). In the dark, the adsorption of U(VI) on MCN*^r* was chemical and monolayer process (Fig. S4 and S5). The adsorption of U (VI) on MCN*^r* increased with increase in *r* from 0.25 to 4.0, which was mainly owing to the increase of the adsorption sites at the edge of pores and $g - C_3N_4$ plates. When the systems were irradiated under visible light, the residual U(VI) in the solutions decreased rapidly. For comparison, the solution without photocatalysts was illuminated under visible light for 40 min, and no significant decline of U(VI) concentration was observed, suggesting that self-photolysis of UO_2^2 ⁺ was negligible. The photocatalytic activity followed the sequence of $MCN_{1.0} > MCN_{2.0} > MCN_{0.5} > MCN_{0.25} > BCN$. By applying the pseudo-first-order reaction $(\text{-}ln(C/C_0) = kt)$ (where *k* is the reaction rate constant (min⁻¹) of U(VI), and C_0 and C are the concentrations of U(VI) (mol/L) corresponding to 0 and *t* min, respectively) [\[52\],](#page-10-12) the corresponding rate constants (k) were deduced as 0.27 (MCN_{1.0}), 0.15 (MCN_{2.0}), 0.08 (MCN_{0.5}), 0.06 (MCN_{0.25}) and 0.04 (BCN) min⁻¹, re-spectively ([Fig. 3B](#page-5-0)). The photocatalytic efficiency of $MCN_{1.0}$ was 6.75 times higher than that of BCN and 100% U(VI) conversion was achieved within 20 min, exhibiting the best performance for uranium extraction. Moreover, the performance of U(VI) reduction over $MCN_{4,0}$ was also compared with that of $MCN_{1,0}$ at a higher initial U(VI) concentration $(2.0 \times 10^{-4} \text{ mol/L})$ (Fig. S7). MCN_{1.0} exhibited much higher efficiency for photocatalytic U(VI) reduction than $MCN_{4.0}$ did (Fig. S7). Due to the largest SSA, the most abundant nanoscale pore walls, and appropriate defects, $MCN_{1.0}$ could achieve optimal surface mass transfer, offering more plentiful active centers and exhibiting the highest visible-light utilization efficiency. Fig. S8 compared the performance difference for the U(VI) reduction between MCN*^r* and reported catalysts. Compared with other photocatalysts, under the studied conditions, $MCN_{1,0}$ hold the optimum photocatalytic efficiency for U(VI) reduction.

[Fig. 3C](#page-5-0) shows the PL spectra with an excitation wavelength of 354 nm. The luminescence was observed over a wide range (400 – 700 nm) for the $MCN_{1.0}$ and BCN samples, and both exhibited a maximum at approximately 470 nm, which was the characteristic bandband PL phenomenon from the n- π^* electronic transitions for g-C₃N₄ [\[25\]](#page-9-13). Clearly, $MCN_{1,0}$ showed lower PL intensity than that of pristine BCN, indicating electron localization on the surface terminal sites [\[38\]](#page-10-2). In other words, the recombination probability of charge carriers in the $MCN_{1.0}$ was lower. As a result, $MCN_{1.0}$ showed higher photocatalytic activity than BCN did, suggesting that the introduction of pore on BCN enhanced the separation and transport efficiency of photo-generated electron-hole (e^-h^+) pairs, which is crucial for improving the photocatalytic reaction. To further understand the recombination kinetics of photoinduced charge, the time-resolved PL spectra of BCN and $\rm{MCN}_{1.0}$ were recorded [\(Fig. 3](#page-5-0)D). The average PL lifetime $(τ)$ was deduced as:

$$
\tau = \frac{A_1 \tau_1^2 + A_2 \tau_2^2 + A_3 \tau_3^2 \cdots}{A_1 \tau_1 + A_2 \tau_2 + A_3 \tau_3 \cdots} \tag{1}
$$

Fitting the decay spectrum revealed three radiative lifetimes with different percentages as given in the inset in [Fig. 3](#page-5-0)D. Specifically, the average PL lifetime of $MCN_{1.0}$ (59 ns) was 1.6 times longer than that of BCN (37 ns). In detail, the percentage of charge carriers with a lifetime shorter than 4 ns decreased from 73.04% for BCN to 66.75% for $MCN_{1.0}$. Meanwhile, the longest lifetime increased from 69.52 ns in BCN to 97.24 ns in $MCN_{1.0}$, and the percentage rose by 0.59%. The longer intrinsic fluorescence lifetime is also expected to improve the transport rate of photocarriers in the MCN*^r* samples [\[53\]](#page-10-13). In summary, the above results indicate that MCN*^r* exhibited a significantly improved photocatalytic performance for the reduction of U(VI), owing to higher

Fig. 3. The variation of UO_2^{2+} concentration vs. illumination time with BCN and MCN_{*r*},
 $C(UO_2^{2+}) = 1.0 \times 10^{-4}$ mol/L, $s/l = 0.5 g/L$, $C({UO_2}^{2+}) = 1.0 \times 10^{-4}$ mol/L, $s/l = 0.5$ g/L, $pH = 6.0$ (A), (Control group: U(VI) was illuminated in the absence of photocatalyst); The corresponding pseudo-first-order rate constant of UO_2^2 ⁺ reduction (B); PL spectra of BCN and $MCN_{1,0}$ under 375 nm excitation (C); Time-resolved PL spectra monitored under 375 nm excitation for BCN (black) and $MCN_{1.0}$ (blue) (D). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

surface area, increased light absorption, and reasonable structural defects. However, when excess amount of templates were applied, the photocatalytic U(VI) reduction was restrained due to the destroy of the mesoporous structures.

3.3. Practical application

As shown in [Fig. 4A](#page-6-0), 73% of the 0.01 mol/L U(VI) could be extracted by photoreduction on $MCN_{1,0}$ within 1 h, and the extraction capacity of uranium reached 2.77 g of U per g-photocatalyst. This high extraction capacity by photocatalytic method was significantly higher than the reported results of physicochemical adsorption and electrochemical methods [\[1,5–9,54\].](#page-9-0) The photoreduction method was also investigated using U(VI)-spiked real seawater (from the Bohai Sea, (118°05′46″, 38°48′02″)) to demonstrate its realistic application. As can be seen from [Fig. 4B](#page-6-0), in the dark, the adsorption of U(VI) was higher than that in pure water owing to higher pH (-8.02) of the seawater. Under visible light irradiation, the kinetic uranium extraction in the seawater was similar to that in pure water, and all U(VI) was extracted within 20 min, indicating a high photoreduction efficiency in a real seawater system.

To evaluate the influence of competing ions (e.g., VO₂⁺, Fe³⁺, Na⁺, K^+ , Cu^{2+} , Zn^{2+} , Ca^{2+} , and Mg^{2+}), the photocatalytic reduction of U (VI) was tested in the presence of the above ions with the same concentration as $U(VI)$ [\(Fig. 4C](#page-6-0)). In the dark, the adsorption of $U(VI)$ was negligibly changed in the presence of the coexisting metal ions. For the competing ions, Na⁺, K⁺, Zn²⁺, Ca²⁺, and Mg²⁺ were almost not adsorbed, while the adsorption of Cu²⁺, Fe³⁺, and VO₂⁺ was 53%, 95%, and 24%, respectively. The different adsorption of coexisting cations is mainly related to their differing valency, radii, softness, and hydrolysis abilities [\[55\].](#page-10-14) After 1 h irradiation, 84.5% of U(VI) was extracted while the extraction efficiency of Cu $^{2+}$ and VO $_2^+$ was 99% and 94%, respectively. However, after irradiation, the removal of Fe significantly declined from 95% to 2.8%. To explore the reaction mechanisms, the valence states of Cu, Fe and V were determined by XPS (Fig. S9). It was confirmed that the binding energy (B.E.) of Cu 2p and Fe 2p shifted lower after irradiation, indicating the photoreduction of Cu(II) and Fe(III) (Figs. S9A and S9D). Combined with Auger Cu LMM spectrum (Fig. S9B), the formation of $Cu₂O$ at the surface of $MCN_{1.0}$ has been confirmed, which resulted in the increased removal of Cu(II) after irradiation. On the contrary, the reduction of Fe(III) to Fe(II) would surely decrease the adsorption affinity of iron ions with the $MCN_{1.0}$ surface. For V, the B.E. of V 2p did not change after irradiation, which indicated that the redox reaction did not occur during irradiation in the case of VO_2^+ . Therefore, the increased removal of V after irradiation might be in terms of the desorption of $Fe³⁺$. During irradiation, the accumulation of Fe(III) on $MCN_{1.0}$ surface significantly decreased and released the active sites that were bounded by VO_2^+ . Overall, in the presence of abundant co-existing ions, despite the deposition of $Cu₂O$ and increased adsorption of V, a relatively high efficiency was still retained for uranium extraction.

The elution of uranium is also crucial to the recovery of uranium and the reuse of photocatalyst. In physicochemical adsorption, uranyl ions adsorbed onto amidoxime polymers should be eluted by strong acidic solution. The elution conditions are generally harsh enough to destroy the sorbent or damage the active sorption sites irreversibly, making the reuse of the sorbents difficult [\[12\].](#page-9-3) For the electrochemical method, both HCl and reverse bias were required to recover 96.2% of the extracted uranium $[1]$. In the present study, after the uranium was photo-deposited, the catalyst was collected and re-dispersed in deionized water. By simply exposing to air, 45% of uranyl ions can be directly recovered, derived from the re-oxidation of reduced uranium in air [\(Fig. 4D](#page-6-0)). Thereafter, when 0.1 M Na₂CO₃ solution was added, 100% U(VI) was recovered ([Fig. 4](#page-6-0)D). This process was efficient, non-polluting, and much easier compared with the physicochemical adsorption and electrochemical methods $[1,13]$. Finally, using the photodeposition-stripping methods, five successive circulating runs were executed to assess the reusability of mesoporous $g - C_3N_4$. As shown in [Fig. 4E](#page-6-0), there was no significant decline in the photocatalytic activity after five recycles, indicating that $MCN_{1.0}$ could maintain stable during the

Fig. 4. Uranium extraction performance using photocatalytic method. $C(U O_2^{2+}) = 0.01$ mol/L, pH = 6.0, s/l = 0.6 g/L (A); U(VI) removal in deionized and spiked real sea water. $C({UO_2}^{2+}) = 1.0 \times 10^{-4}$ mol/L, pH = 6.0, s/l = 0.5 g/L (B); Removal of metal ions with MCN_{1.0} before and after irradiation under visible light for 1 h. *C*(metal ions) = 1.0 × 10−4 mol/L, pH = 6.0, *s*/*l* = 0.5 g/L (C); The elution of photo-induced uranium deposits (D); Cycling runs for the photocatalytic reduction of UO2²⁺ over MCN_{1.0} sample under visible light irradiation (E); XRD patterns before and after irradiation (F). *C*(UO2²⁺) = 1.0 × 10^{−4} mol/L, pH = 6.0, *s/* $l = 0.5$ g/L.

photocatalytic reactions. This was further supported by the XRD and FT-IR investigations, where no distinct alteration was observed for structure and functional groups on $MCN_{1,0}$ after irradiation ([Fig. 4](#page-6-0)F and Fig. S10A). The mesoporosity of $MCN_{1.0}$ was also well maintained after five cycle tests (Fig. S10B). The above results proved that the $MCN_{1.0}$ catalyst exhibits high efficiency, appropriate selectivity, good stability and reusability for the photocatalytic removal of U(VI) and is promising for practical application in uranium extraction from aqueous solution.

3.4. Photocatalytic reduction mechanism

Although it has been widely accepted that U(VI) was photoreduced to U(IV), the transformation process still remains controversial, meriting further research. As shown in Fig. S11, the color of U(VI)-adsorbed $MCN_{1.0}$ is yellow, which turned to gray under light irradiation. Salomone et al. [\[52\]](#page-10-12) demonstrated that the dark gray precipitation formed by TiO₂-induced photoreduction was UO_{2+x} (x = 0–0.25). Similarly, Amadelli et al. [\[56\]](#page-10-15) reported uranyl oxide photodeposited on TiO₂ powder with a stoichiometry close to U_3O_8 by infrared analyses. In addition, Lee et al. [\[3\]](#page-9-1) observed three lower oxidation states (i.e., U^{4+} , $U^{14/3+}$, and $U^{16/3+}$) in the production on semiconductor electrodes. XPS identified the major ionic species after photoelectrochemistry treatment as U^{6+} , $U^{4.6+}$, and U^{4+} [\[57\].](#page-10-16) Meanwhile, other studies confirmed the copresence of both U(VI) and U(IV) after photoreduction [\[24,25,58\].](#page-9-9)

[Fig. 5A](#page-7-0) shows the XRD patterns of $MCN_{1,0}$ before and after U(VI) photoreduction. One can see that for U(VI)-adsorbed $MCN_{1.0}$, no new peak was observed, except for the characteristic peaks from $MCN_{1.0}$ $(-13.1°$ and 27.4°). However, the typical peaks of UO₂ located at \sim 28.1°, 32.4°, 46.8° and 55.9° were observed after visible light irradiation, indicating that the conversion of $U(VI)$ to $UO₂(s)$, which is similar to the results in Li et al. [\[13\]](#page-9-4). It was also very interesting to note that meta-schoepite $((UO_2)_8O_2(OH)_{12}(H_2O)_{10}$ or $UO_3·2H_2O$) was formed at pH \sim 6.0, which could be further reduced to form UO₂(s)

([Fig. 5A](#page-7-0)). This indicated that although U(VI) was partly precipitated under high pH conditions, all U(VI) species could be reduced to $UO₂$ by $MCN_{1.0}$ via photocatalysis [\[13\]](#page-9-4).

The aforementioned results were further supported by the morphology observation. At pH \sim 5.0, no distinct change on MCN_{1.0} was observed after U(VI) sorption. While after light irradiation, nanosized particles formed on the surface of $MCN_{1.0}$ ([Fig. 6](#page-7-1)A). The interference fringe from the HRTEM image ([Fig. 6B](#page-7-1)) showed a spacing of 0.310 nm, which is in accordance with the interplanar distance of the $(1\ 1\ 1)$ plane of the UO₂ standard (0.315 nm, 00-041-1422-ICDD) [\[59\]](#page-10-17). Under high pH (6.0) conditions, needle-like meta-schoepite particles were formed on the $MCN_{1.0}$ surface (Fig. S12). These needle-like minerals gradually vanished after visible light irradiation, and $UO₂$ nanoparticles were formed on the $MCN_{1.0}$ surface (Fig. S12E and S12F).

[Fig. 5](#page-7-0)B shows the U 4f spectra of U(VI)-accumulated $MCN_{1.0}$ before and after irradiation. Before irradiation, U $4f_{7/2}$ and U $4f_{5/2}$ peaks of the U(VI)-adsorbed $MCN_{1,0}$ surface were centered at 381.7 and 392.9 eV, respectively. However, the B.E. of U 4f shifted lower after visible light irradiation. U 4f signal can be quantitatively divided into two components, ascribing to the characteristics of U(VI) (at \sim 380.9 and \sim 391.8 eV) and U(IV) (at \sim 380.0 and \sim 390.9 eV) species, respectively [\[13\]](#page-9-4). According to the XPS analyses, 41.7% of U(IV) and 58.3% of U (VI) were detected, and the peak position of U(VI) displayed a significant shift compared to UO_2^{2+} [\(Fig. 5](#page-7-0)B), possibly due to the surface re-oxidation of the generated $UO₂$ by photo-hole or oxyhydrogen radicals [\[13\]](#page-9-4). Therefore, X-ray absorption fine structure (XAFS) technique was further applied to determine the species of deposited uranium. [Fig. 5C](#page-7-0) shows the U L_{III}-edge XANES of the uranium standard $(UO_2^2$ ⁺, UO₂(s), U₃O₈(s)) and uranium-loaded MCN_{1.0} after irradiation. Obviously, the absorption edges of standard $UO_2(s)$ and $U_3O_8(s)$ shifted to lower energies compared to $\mathrm{UO_2}^{2+}$. In U $\mathrm{L_{III}}$ -absorption, the near edges and shape of the uranium-loaded $MCN_{1.0}$ after irradiation showed perfect agreement with the $UO₂$ standard ([Fig. 5C](#page-7-0)), which confirmed the predominant U(IV) species. [Fig. 5D](#page-7-0) and 5E show the k^3 -weighted

Fig. 5. XRD patterns of MCN1.0 before and after photoreduction with 5.0 × 10−4 mol/L U(VI) at pH 5.0 and 6.0 (A); XPS spectra of U 4f before and after irradiation (B); The U L_{III}-edge XANES spectra of U(VI)-loaded MCN_{1.0} after irradiation (C); Uranium L_{III}-edge k³-weighted EXAFS spectra (D) and the corresponding FTs (E) for the reference samples and U(VI)-reacted MCN_{1.0} (a: U(VI) reacted MCN_{1.0}; b: Nano-uraninite standard (Data from Fletcher et al. [\[60\]](#page-10-18)); c: UO₂ standard (Data from Fletcher et al. $[60]$); d: UO_2^2 ⁺ standard).

χ(*k*) function and corresponding Fourier transforms (FTs) of the standards (UO₂(s) and UO₂²⁺), as well as the U-loaded MCN_{1.0} samples after irradiation. After visible light irradiation, the *k*³ *χ*(*k*) function exhibited similar oscillation features to that of reference $UO₂$. The fitting results are tabulated in [Table 2.](#page-8-1) For the typical structure of uranyl, the spectral fits lead to \sim 2.0 O_{ax} (axial oxygen atom) at 1.8 Å and \sim 5.9 O_{eq} (equator oxygen atom) at 2.5 Å, which are in agreement with the distances reported for other uranyl aqueous species [\[43,60,61\].](#page-10-5) While after irradiation, the central U atom was surrounded by \sim 6.10 atoms at $R \sim 2.4$ Å and \sim 14.40 atoms at $R \sim 4.5$ Å, as well as \sim 3.1 U atoms at R \sim 3.9 Å, respectively, which are ascribed to the near-neighbor U-O₁

shell (the first oxygen shell), U-U shell, and outer U-O₂ shell (the second oxygen shell) of $UO₂$ ([Table 2\)](#page-8-1) [\[60,61\].](#page-10-18) The result indicated that the phase of $UO₂(s)$ was formed after visible light irradiation. In addition, EXAFS fits clearly showed an additional coordination shell at \sim 2.1 Å with *CN* ~2.5 O atoms in the first shell. Clearly, it was not the U-O interatomic distance from UO_2 or UO_2^{2+} but may arise from other uranium oxides owing to the re-oxidization of $UO₂$ surface, in accordance with XPS results. Apparently, the detected species was document as $UO₂$ in EXAFS, which is significantly higher than that in XPS (41.7% U(IV)). The difference in the detected U^{4+} content may be mainly caused by the difference in accuracy between XPS and EXAFS

Fig. 6. TEM images of U(IV) deposition on MCN_{1.0} (A); HRTEM of U(IV) deposition on MCN_{1.0} (B) and Corresponding FFT image (inset); The image of U(IV) deposition on MCN_{1.0} (C); Elemental distribution maps of C (D), N (E), and U (F) after irradiation. $C(UO_2^{2+}) = 5.0 \times 10^{-4}$ mol/L, pH = 6.0.

Table 2

Numerical results from modeling of the EXAFS data from the standards and the uranium-loaded $MCN_{1.0}$ samples.

Note: R, bond distance; CN, coordination number; σ^2 , Debye-Waller factor. The standard data of UO₂ and nano-uraninite (U^{IV}O₂(s)) were derived from Fletcher et al. [\[60\]](#page-10-18).

(XPS technique only characterizes the top layer within 10 nm of the samples, while EXAFS could record the bulk information of the samples) [\[13,62\]](#page-9-4).

The effect of holes and radicals on U(VI) photocatalytic reduction over MCN*^r* were examined to further elucidate the reaction mechanism. ESR spin-trap experiment was carried out to detect the generation of free radicals ('OH and 'O₂⁻) in MCN_{1.0} system under visible light condition. As shown in [Fig. 7](#page-8-2)A, there was no ESR signal when the reaction was performed in the dark. While under visible light irradiation, the characteristic peaks of DMPO-`O₂⁻ and DMPO-`OH (1:2:2:1 quartet pattern) adducts could be observed [\[63\],](#page-10-19) and the intensity gradually increased with the irradiation time. These results indicated that ${^{\cdot}O_2}^$ and 'OH indeed formed during the photocatalytic process. To test the efforts of these radicals on the photocatalytic reduction of U(VI), quenching experiments were further conducted. TBA (100 mmol/L), P-BQ (1 mmol/L) and methanol (V(CH₃OH/H₂O) = 1:30) were added as scavengers for `OH radicals, `O₂⁻ radicals and holes, respectively. From [Fig. 7B](#page-8-2), methanol could act as hole trap for enhancing the separation of photoinduced electrons and holes, thus to enhance the photocatalytic reduction of U(VI). Moreover, during methanol oxidization process, reductive free radicals (' CO_2 ⁻) might be produced to contribute to the reduction of U(VI) $[64]$ (eqns. (3) and (4)). P-BQ could

remarkably inhibited U(VI) removal, demonstrating the generated ${^\cdot\mathrm{O_2}}^$ radicals in the system MCN*r*/U(VI)/visible light was also conductive to the photocatalytic reduction of U(VI) $[57,65]$. While the presence of TBA have slight influence on the U(VI) reduction, indicating that 'OH radicals played a negligible role in the reaction process. The above results confirmed that both photoinduced electrons and ${^\cdot O_2}^-$ radicals were involved in the photocatalytic reduction of U(VI).

In summary, the photocatalytic reduction process is illustrated as follows. Under visible light irradiation, U(VI) was reduced to form $UO₂$ deposits by photogenerated electron and $\, \dot{O}_2 \, \bar{\,}$. For the formed metaschoepite particles under high pH conditions, there are two possible transformation mechanisms. One is that the meta-schoepite precipitates on the surface directly obtained photo-electrons or ${^{\text{·}}O_2}^-$ to form UO_2 The other is that upon light irradiation, soluble U(VI) was converted to $UO₂$ on solid surface, breaking the chemical equilibrium between metaschoepite precipitates and soluble U(VI). With the decrease in soluble U (VI), meta-schoepite gradually re-dissolved into solution, and further reduced to form $UO₂$. The proposed processes for the photocatalytic reduction of U(VI) to U(IV) is exhibited via Eqs. $(2)-(12)$:

 $MCN + h\nu \rightarrow e_{(MCN)}^{-} + h_{(MCN)}^{+}$ (excitation) (2)

$$
CH_3OH + 6OH^- + 6h^+ \rightarrow CO_2 + 5H_2O
$$
 (3)

$$
CH_3OH + OH^- + 4h^+ \rightarrow CO_2^- + 5H^+ \tag{4}
$$

$$
O_2 + e^- \rightarrow O_2 \tag{5}
$$

For soluble UO_2^{2+} :

$$
O_2^{2+} + 2e^- \to UO_2(s) \tag{6}
$$

$$
UO_2^{2+} + 2^0O_2^{-} \rightarrow UO_2(s) + 2O_2
$$
 (7)

For formed UO_3 :2H₂O precipitates, two ways are presumed:

$$
UO_3:2H_2O + 2H^+ + 6e^- \to UO_2(s) + 3H_2O
$$
 (8)

$$
UO_3:2H_2O + 2H^+ + 2O_2^- \rightarrow UO_2(s) + 2O_2 + 3H_2O
$$
 (9)

or

 U

$$
UO_3:2H_2O \to UO_2^{2+} + H_2O + 2OH^-
$$
 (10)

$$
UO_2^{2+} + 2e^- \to UO_2(s) \tag{11}
$$

$$
UO_2^{2+} + 2^1O_2^{-} \rightarrow UO_2(s) + 2O_2
$$
\n(12)

Fig. 7. ESR spectra of radical adducts trapped by DMPO (`O₂ $^-$ and `OH) in MCN_{1.0} dispersion in the dark and with the visible light irradiation for 5 min, 10 min and 15 min (A); The photocatalytic reduction plots of U(VI) over MCN_{1.0} with the addition of hole, O_2^- and 'OH radical scavenger under visible light irradiation (B), Note: only $U(VI)$ and $MCN_{1.0}$ were contained in the control group.

4. Conclusions

Mesoporous $g - C_3N_4$ was fabricated for the photocatalytic reduction of U(VI) under visible light. Owing to the introduction of adjustable meso-holes, MCN*^r* provided a high surface-to-volume ratio, appropriate surface deficiency, and enhanced visible light absorption. $MCN_{1.0}$ possessed the highest surface area (191 m^2/g , increased by 21.8 fold) and largest pore volumes $(52.1 \text{ cm}^3/\text{g}$, increased by 13 fold), exhibiting the highest performance for photocatalytic U(VI) reduction (6.75 fold higher than BCN). A high extraction capacity of uranium (2.99 g/g) was achieved over light illuminated $MCN_{1.0}$. In the presence of co-existing ions, $MCN_{1.0}$ exhibited relatively high selectivity for the photo-reduction of U(VI). The deposited uranium on $MCN_{1.0}$ could be efficiently recovered simply by exposure in air for 24 h, and further eluted with a 0.1 mol/L Na₂CO₃ solution. Moreover, MCN_r could maintain stable during the photocatalytic reactions. XRD and TEM confirmed that U(VI) was converted to $UO_2(s)$ phase on the MCN_r surface during photocatalysis, while XPS, XANES, and EXAFS further confirmed that the formed $UO₂(s)$ on MCN_r surface was partly re-oxidized by radicals and holes. However, the process of re-oxidation of $UO₂(s)$ did not lead a dissolution of deposited uranium, and thus, exerted no obvious influence on the extraction efficiency of U(VI). Above all, MCN*^r* here represent potentially a simple and cost-effective photocatalytic materials, showing clear advantages over traditional uranium sorbent materials. This strategy can provide guidance toward efficient, economic, and applicable treatment of uranium from water.

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Conflicts of interest statement

The authors declare no competing financial interests.

Appendix A. Supplementary data

Supplementary data to this article can be found online at [https://](https://doi.org/10.1016/j.cej.2019.123193) [doi.org/10.1016/j.cej.2019.123193.](https://doi.org/10.1016/j.cej.2019.123193)

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